

# EXPERIMENTAL OBSERVATION OF MOVING DISCRETE BREATHERS IN GERMANIUM

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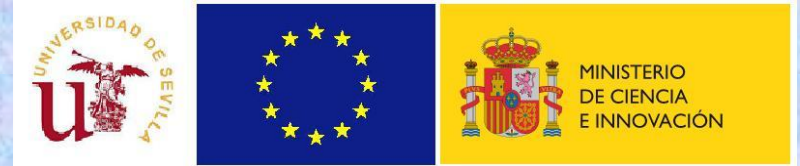
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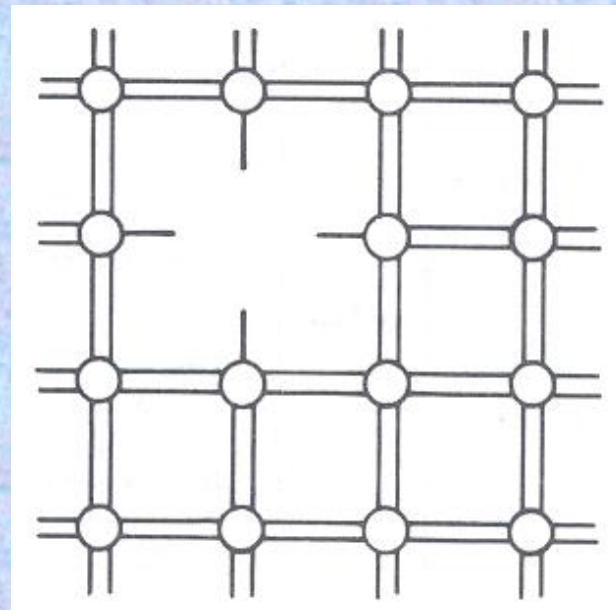
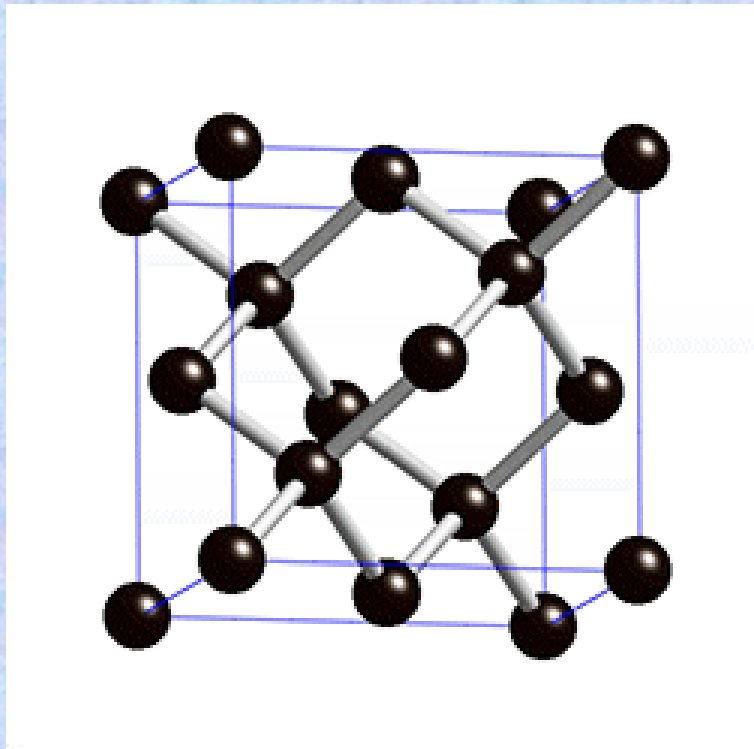
**Conference Quodons in Mica, Altea 2013: A meeting in honour of Mike Russell**

Nonlinear localized travelling excitations in crystals.

**Altea, Spain, September 18-21, 2013**

# Germanium

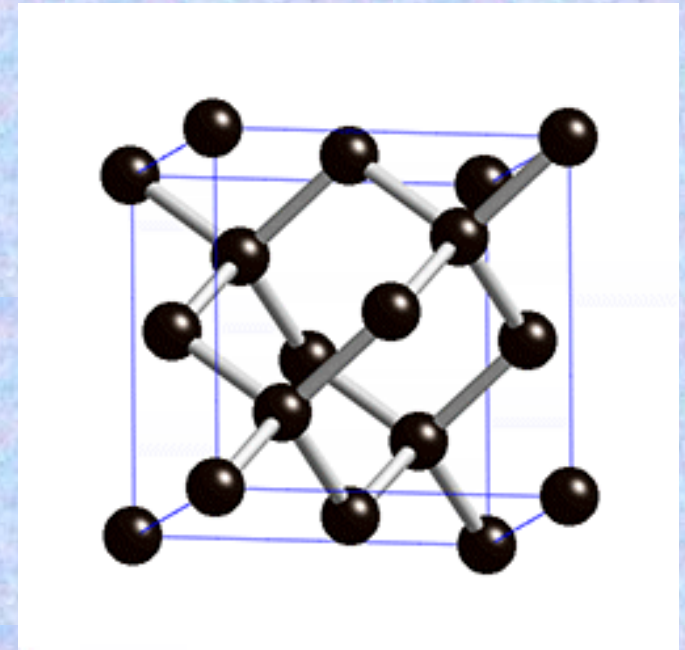
- Unlikely structure for moving discrete breathers



# Outline

We believe that there are moving discrete breathers in Ge

- What property allows for their detection? Defect annealing
- What technique we use to detect them: DLTS
- How efficient DB with respect to phonons for annealing?
- What characteristics about breathers in Ge we can estimate?



# Germanium

At room temperature (RT, 300 K) the size of the band gap is 0.66 eV while increasing with decreasing temperature to 0.74 eV at a temperature of a few K

Relative effective masses (density of states)

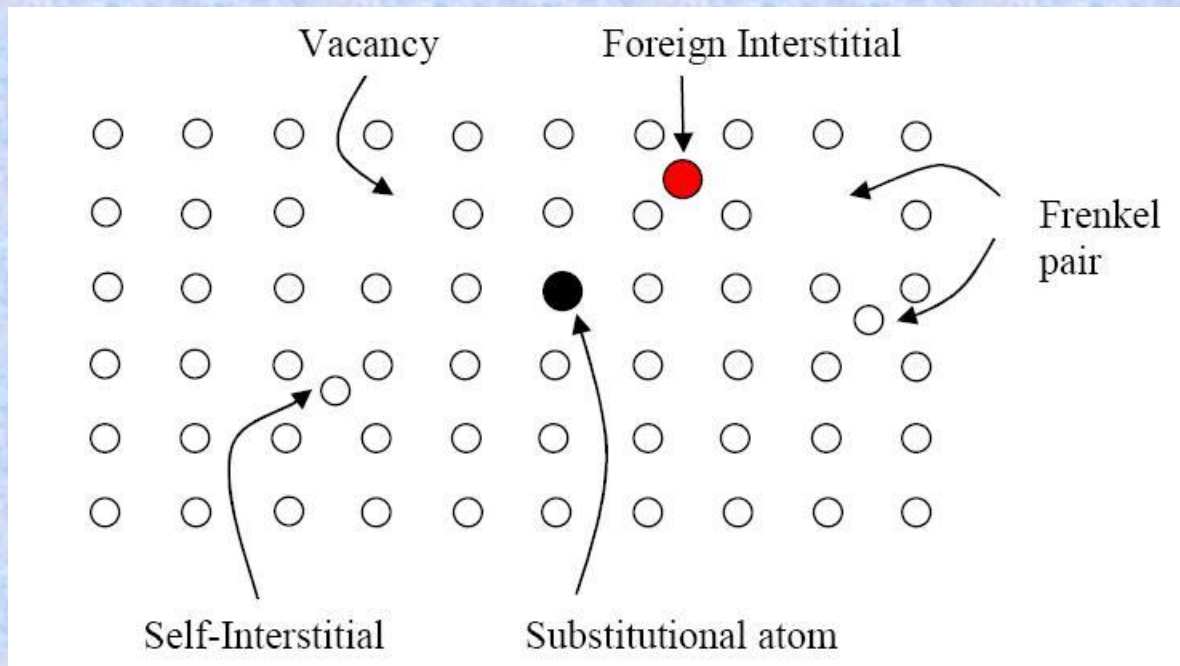
|          | Si   | Ge   |
|----------|------|------|
| Electron | 1.08 | 0.55 |
| Hole     | 0.56 | 0.37 |

Typical (low doping) values

|   | Si  | Ge        |
|---|-----|-----------|
| Electron mobility ( $\text{cm}^2/\text{Vs}$ ) |     | 1350 3900 |
| Hole mobility ( $\text{cm}^2/\text{Vs}$ )     | 480 | 1900      |

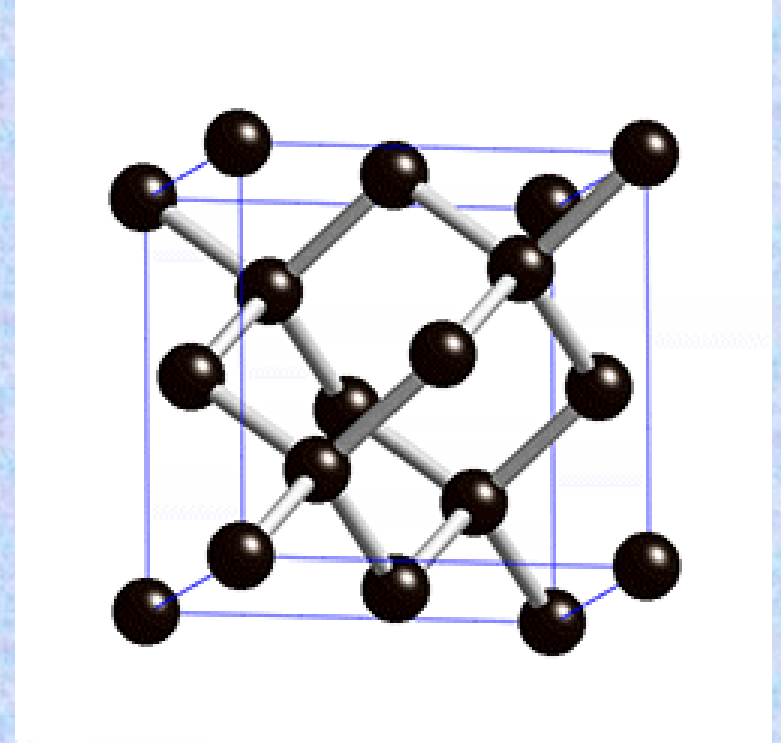
# Defects in germanium

- Can be produced by irradiation
- Of technological interest

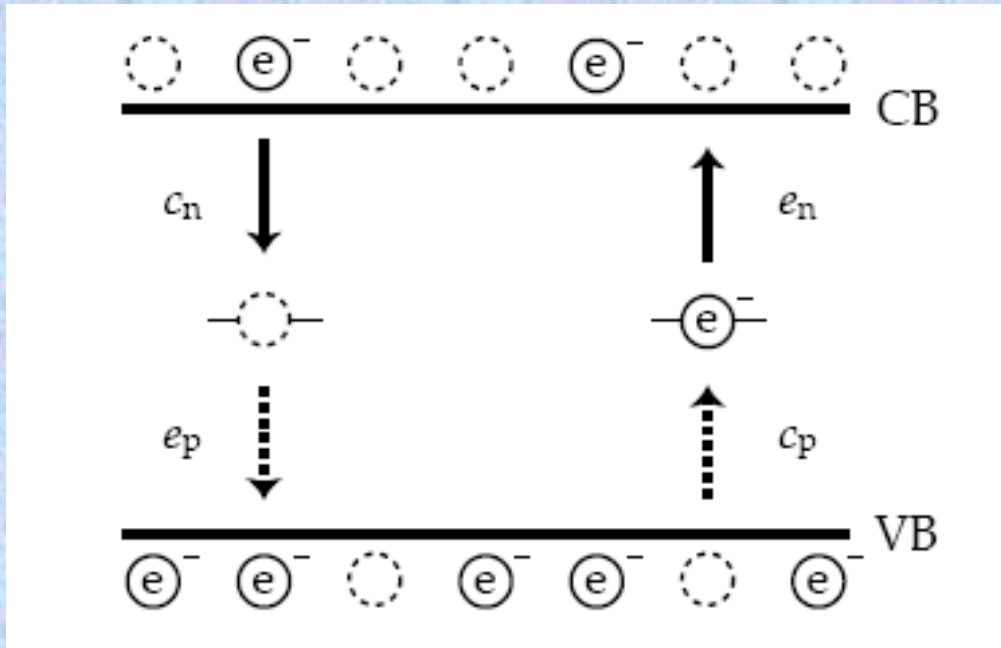


# Some complex defects in germanium

- Di-vacancy V-V
- Tri-vacancy  $V_3$ , Tetra-vacancy  $V_4$
- Vacancy-Hydrogen  $VH_n$
- $I_2, I_3, \dots$
- A center:  
Foreign interstitial 0-Vacancy
- E center  
Substitutional atom-Vacancy  
Sb doped Ge: Sb-V



# Defect as electron and hole traps



Dynamics of an energy state.

The solid arrows :  
electron capture/emission

Dashed arrows:  
hole capture/emission.

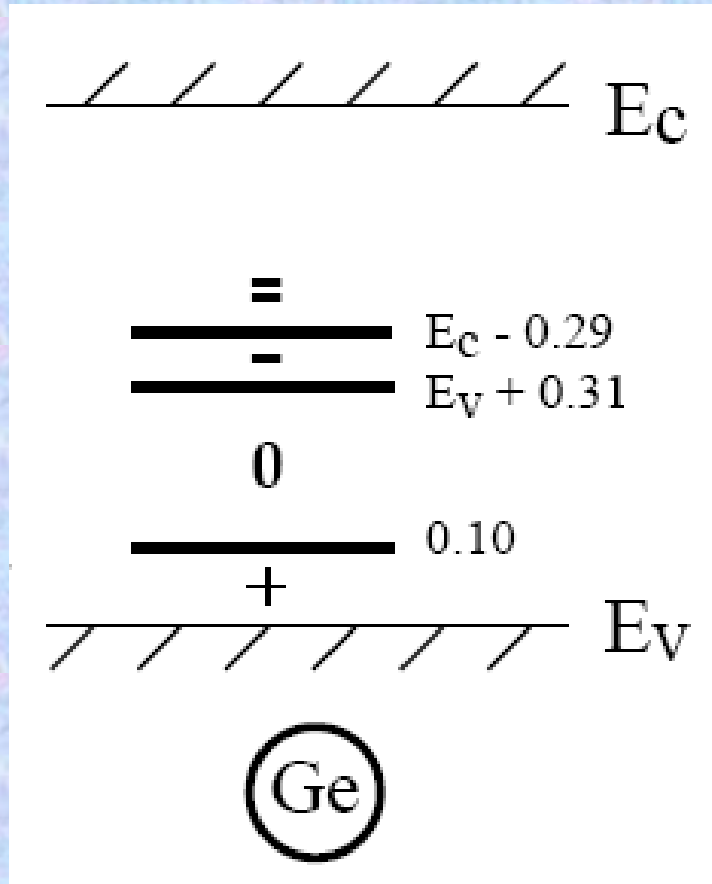
Empty level. defect  
can  
-Capture an electron  
from the CB ( $c_n$ )

Defect populated by an  
electron can :  
-Emit an electron  
to the CB ( $e_n$ )

-Emit a hole  
to the VB ( $e_p$ )

- Capture a hole  
from the VB ( $c_p$ )

# Electron and hole traps



- Defects have acceptor or donor levels
- Deep levels: far from the nearest band ( $>0.1$  eV).
- $E_c - E_v = 0.67$  eV
- Electron trap:
  - n-type  $E_{0.29}$
- Hole trap
  - p-type  $H_{0.10}$



# Capture and emission rate of an electron trap

Capture rate  $c_n = \sigma_n \langle v_n \rangle n$

Emission rate  $e_n = \sigma_n \langle v_n \rangle N_c \exp\left(-\frac{E_T}{kT}\right) = \sigma_n \gamma_n T^2 \exp\left(-\frac{E_T}{kT}\right)$

Activation energy for electron emission:  $E_T = E_C - E_t$

$\gamma_n$  depends on fundamental constants and  $m_e^*$

Signature of a defect:  $E_T$ ,  $\sigma_n$

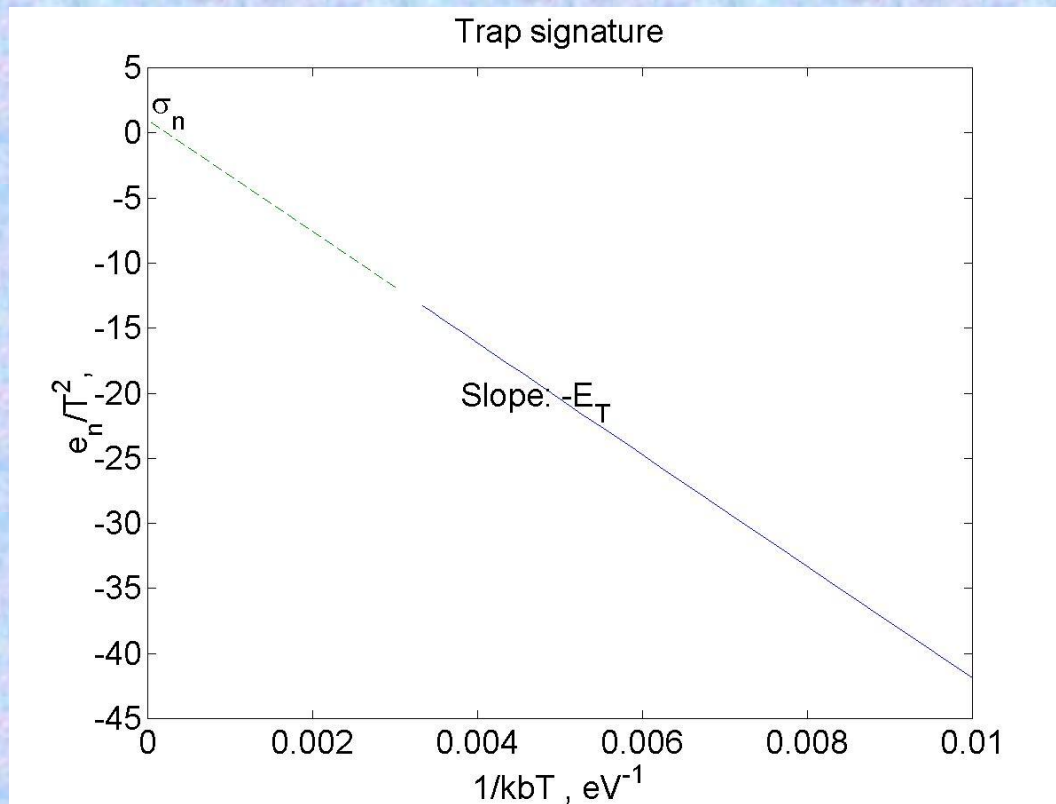
Number of defects or traps:  $N_T$

Other type of parameters:

- Energy barrier for annealing:  $E_0$
- Temperature of annealing:

# Capture and emission rate of an electron trap

$$e_n = \sigma_n \langle v_n \rangle N_c \exp\left(-\frac{E_T}{kT}\right) = \sigma_n \gamma_n T^2 \exp\left(-\frac{E_T}{kT}\right)$$



# DLTS: Deep Level Transient Spectroscopy

**Objective:** find the trap parameters and concentration

**Magnitude measured:** electron lifetime in a trap: at different temperatures

$$\tau_n = 1/e_n ; \tau_n = \tau_n(T)$$

Needs a depletion layer as in p-n junction or Schottky diode under reverse bias

Procedure: Fill all the trap levels and measures the capacitance at two different times as traps emit electrons and discharge.

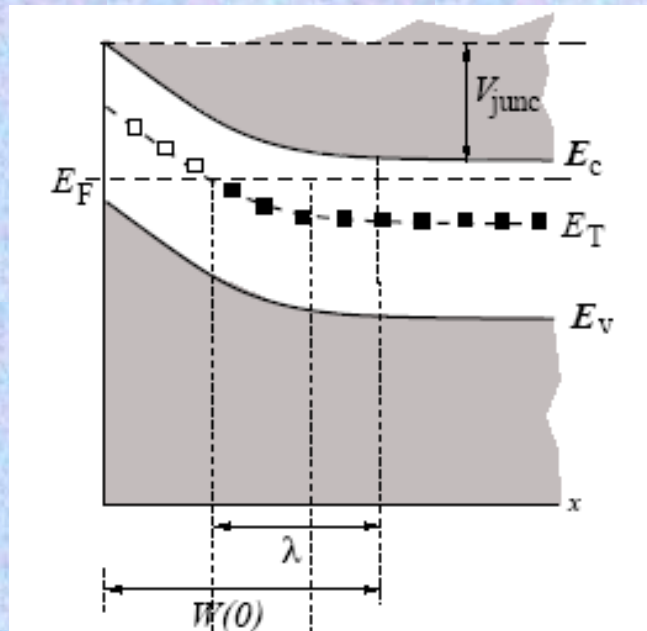
$$\Delta C = C(t_1) - C(t_2)$$

DLTS signal or transient:

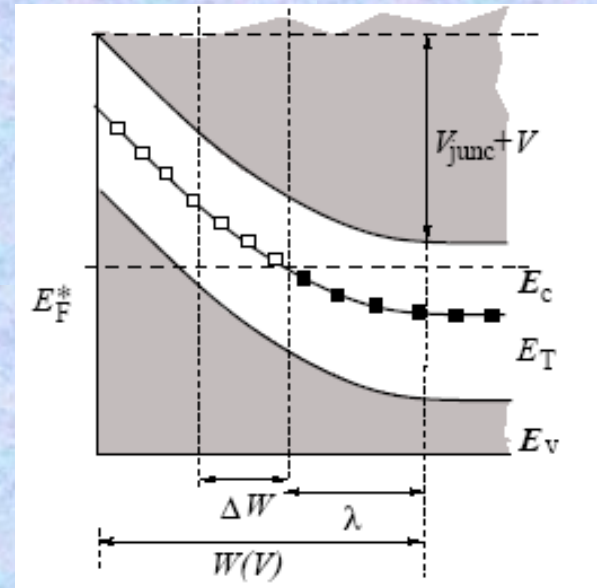
Repeats the procedure as the temperature changes.  $\Delta C = \Delta C(T)$

# DLTS: Deep Level Transient Spectroscopy

p+-n junction or Schottky diode (metal-semiconductor)



Zero bias



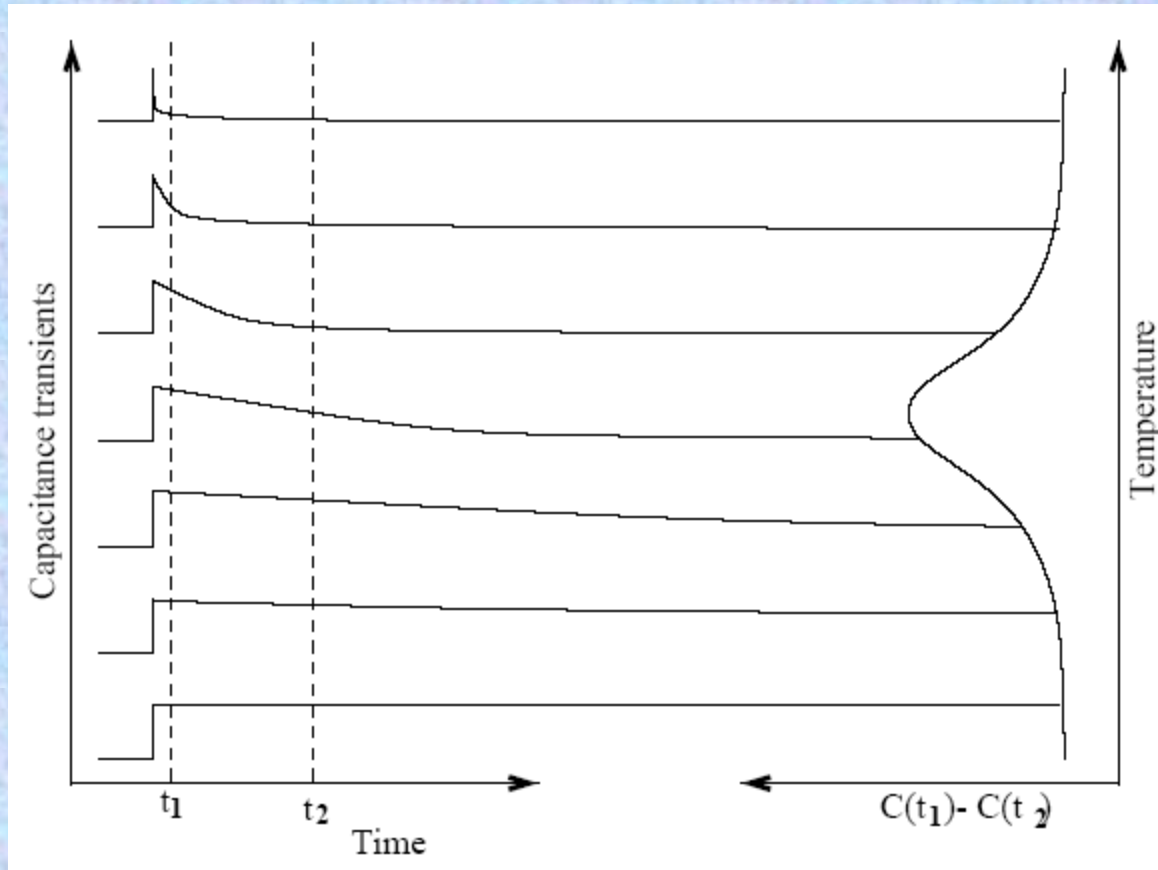
Reverse bias

$\Delta W$ : depletion region.

Then: biasing pulse fills the depletion region with electrons filling the traps, which subsequently emit

# DLTS: Deep Level Transient Spectroscopy

Capacitance transients:



Zero bias

$$C(t) = C_0 \exp(-t / \tau_n)$$

$$\tau_n = 1/e_n$$

Life time of an electron  
in the trap decreases  
with T

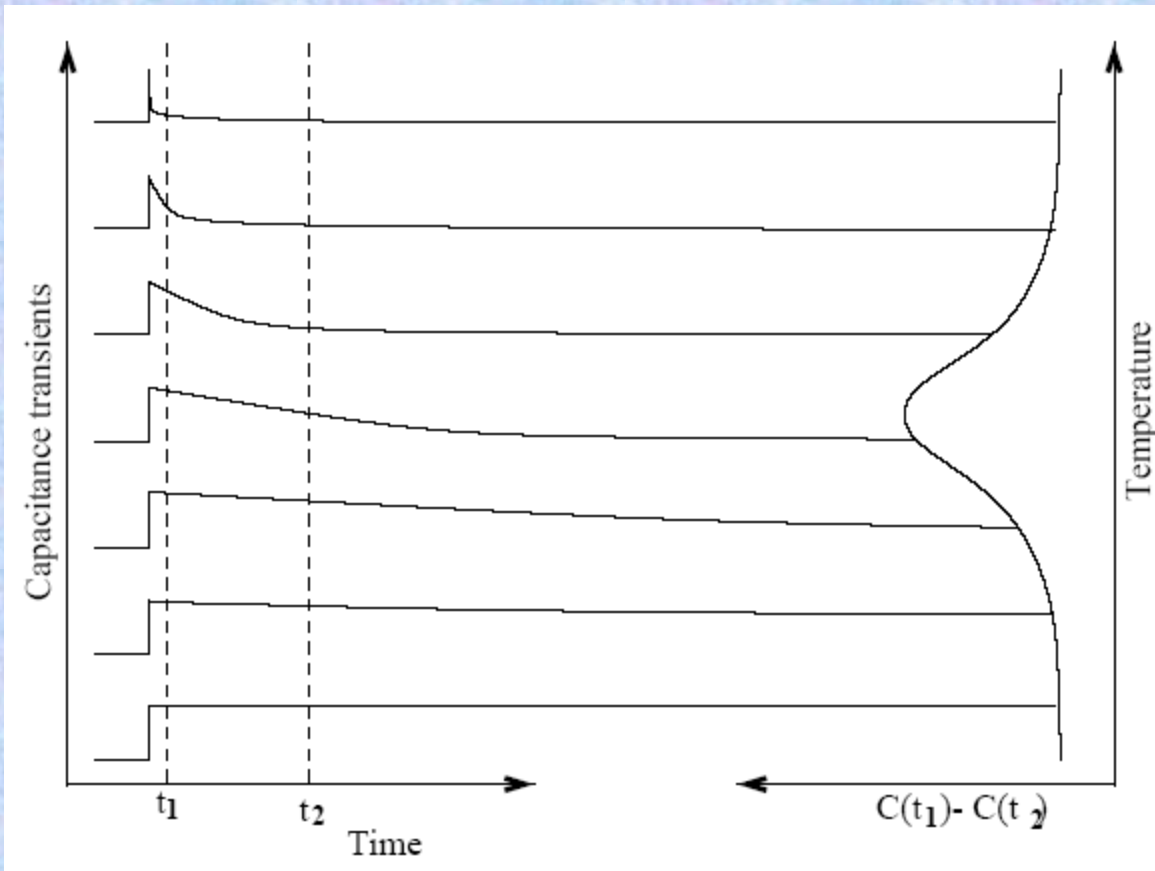
# DLTS: Rate window

$\Delta C(\tau) = C_0(\exp(-t_1 / \tau) - \exp(-t_2 / \tau))$  has a maximum at

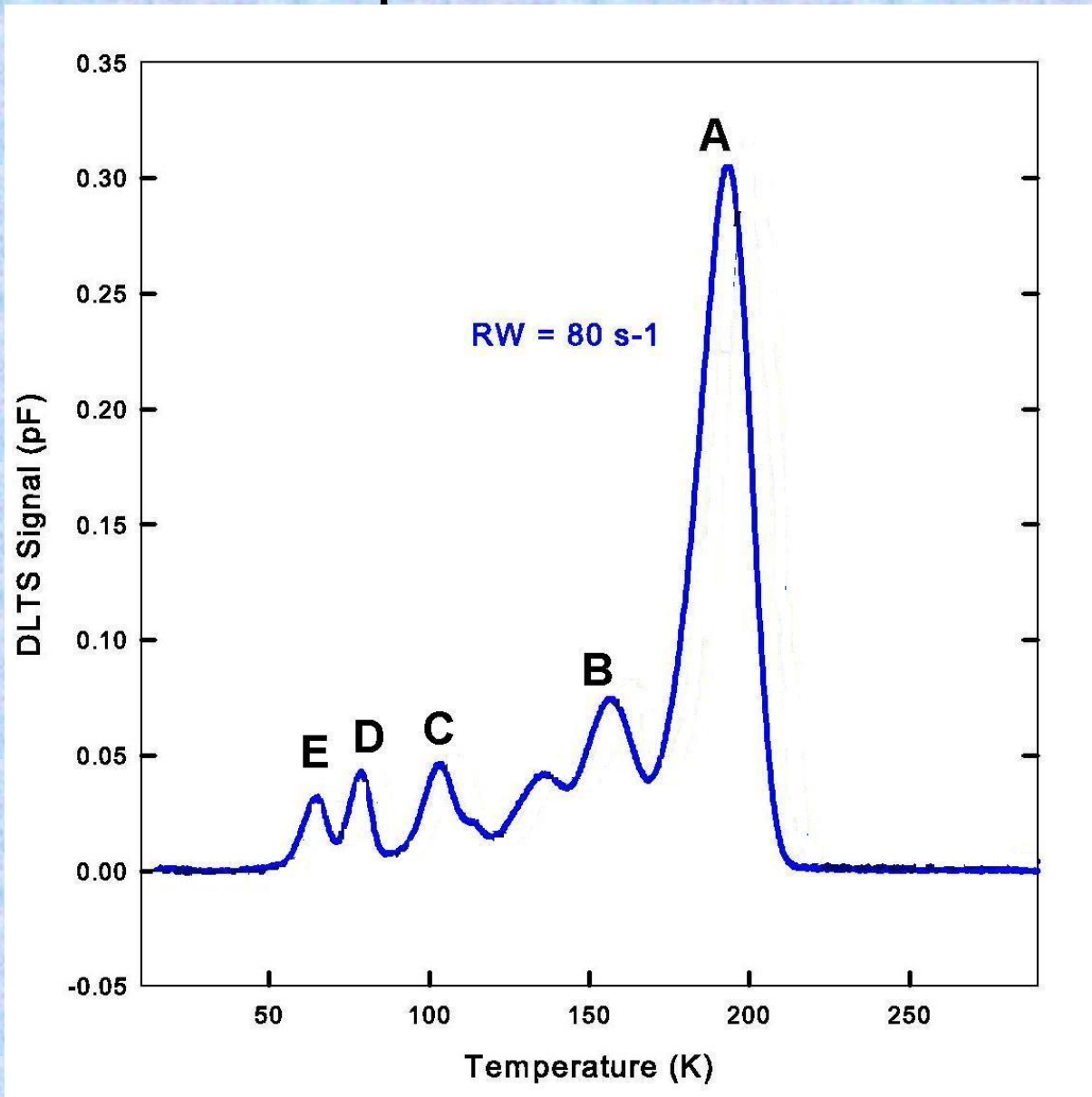
$$RW = 1 / \tau_{\max} = \ln(t_1 / t_2) / (t_1 - t_2)$$

Typical RW are  
80 s<sup>-1</sup> and 200 s<sup>-1</sup>

When the emission rate  
 $e_n(T) = RW$  there is a  
maximum at  $\Delta C = \Delta C(T)$



# DLTS: Example



Defect A:

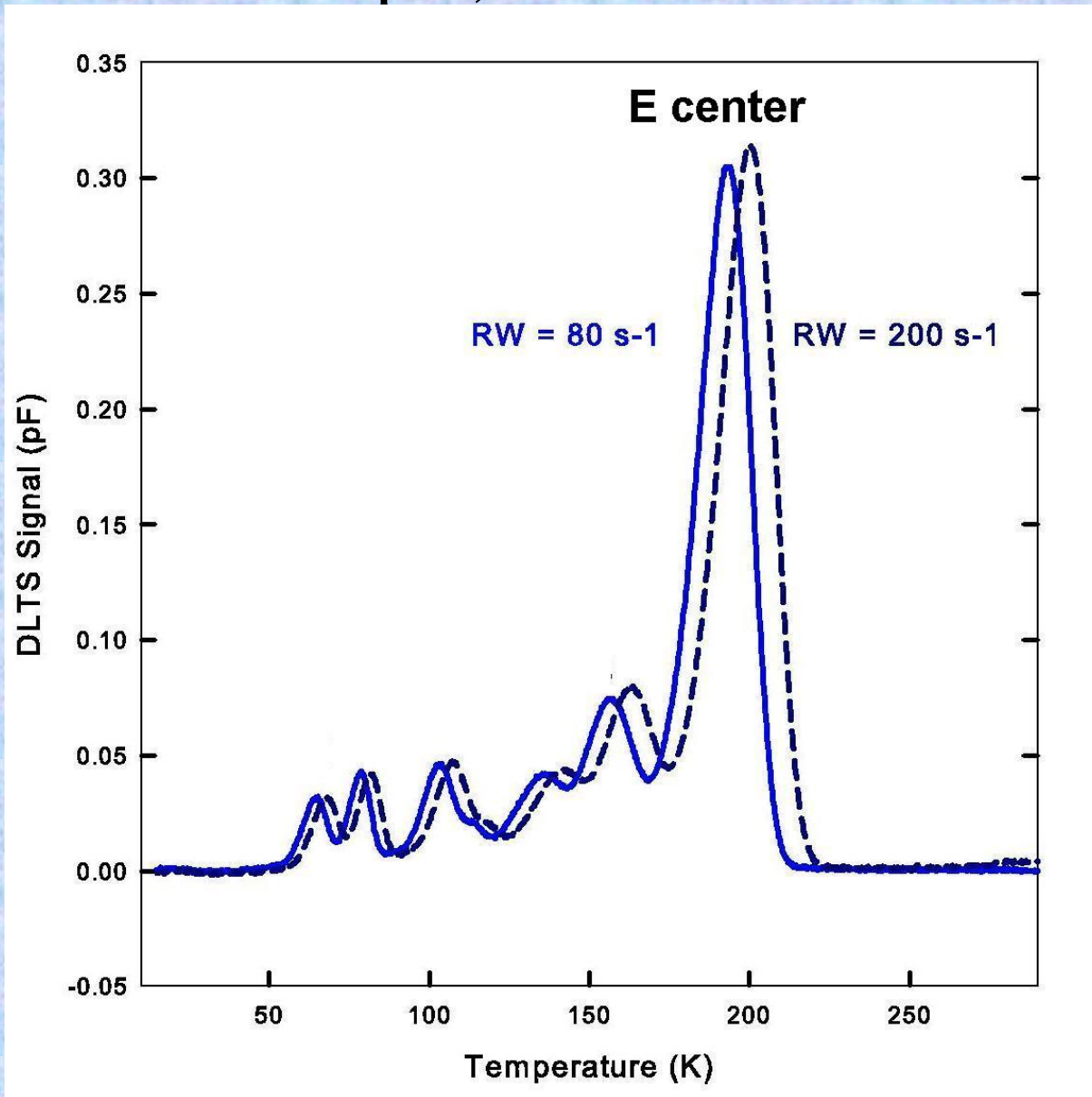
$$RW_1 = 80s^{-1} = e_n(T_A)$$

Defect B:

$$RW_1 = 80s^{-1} = e_n(T_B)$$

Defect C .....

# DLTS: Example, two RW: Defect E center



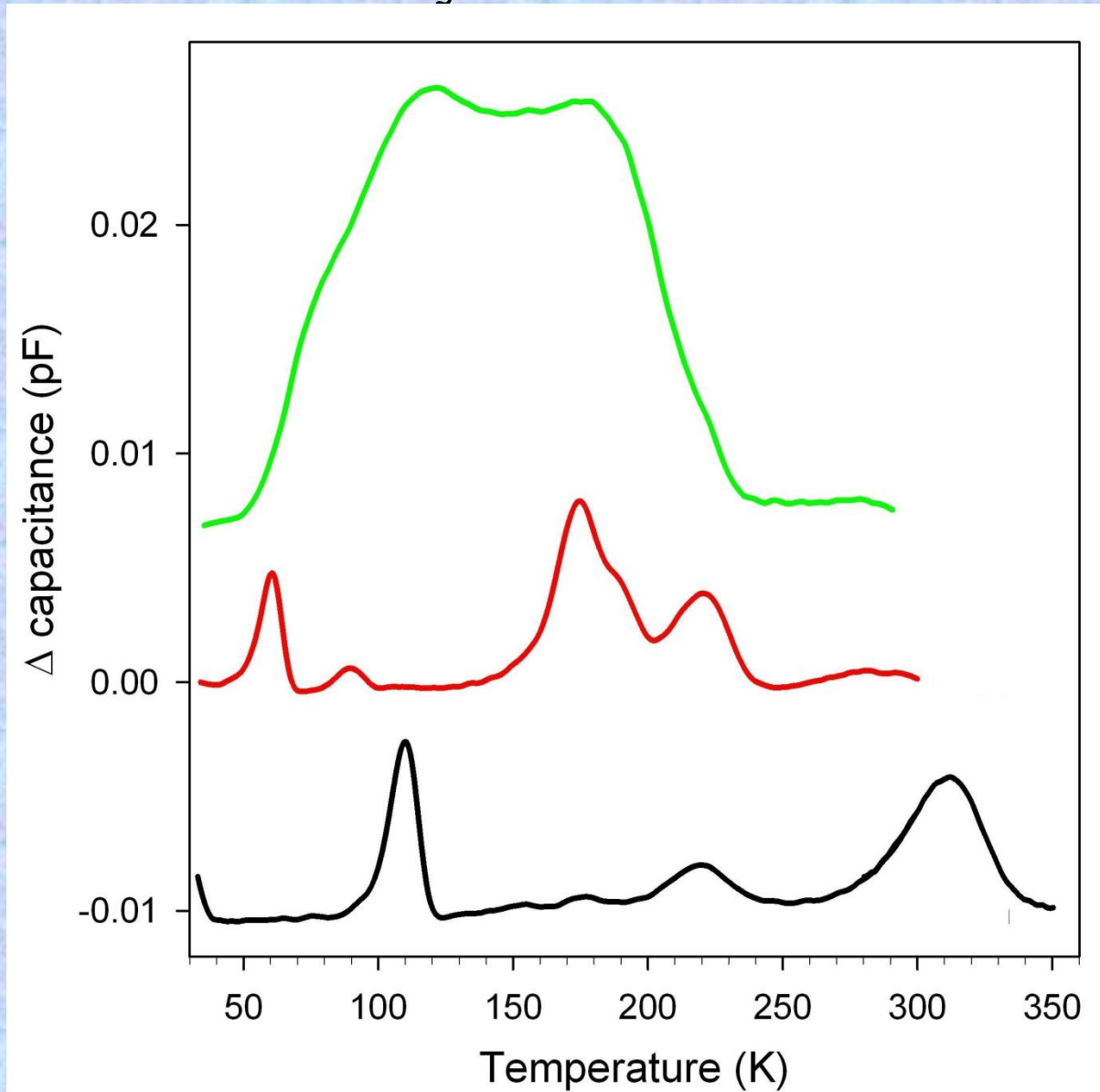
$$RW_1 = 80s^{-1} = e_n(T_1) = \sigma_n \gamma_n T_1^2 \exp\left(-\frac{E_T}{kT_1}\right)$$

$$RW_2 = 200s^{-1} = e_n(T_2) = \sigma_n \gamma_n T_2^2 \exp\left(-\frac{E_T}{kT_2}\right)$$

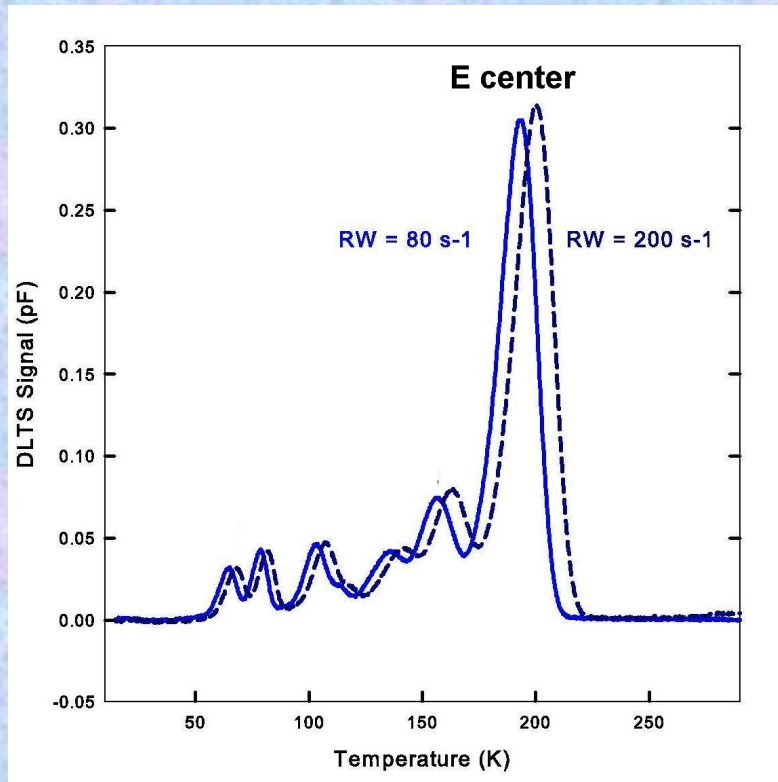
$\sigma_n$  and  $E_T$  can be determined



# DLTS: Two many defects!



# Number of traps for n-type semiconductor:



$N_D$  : concentration of donors

$$\frac{\Delta C}{C_0} = \frac{N_t}{2N_D}$$

The number of traps  $N_t$  can be determined

The depletion layer with  $W$  increases with the bias potential:

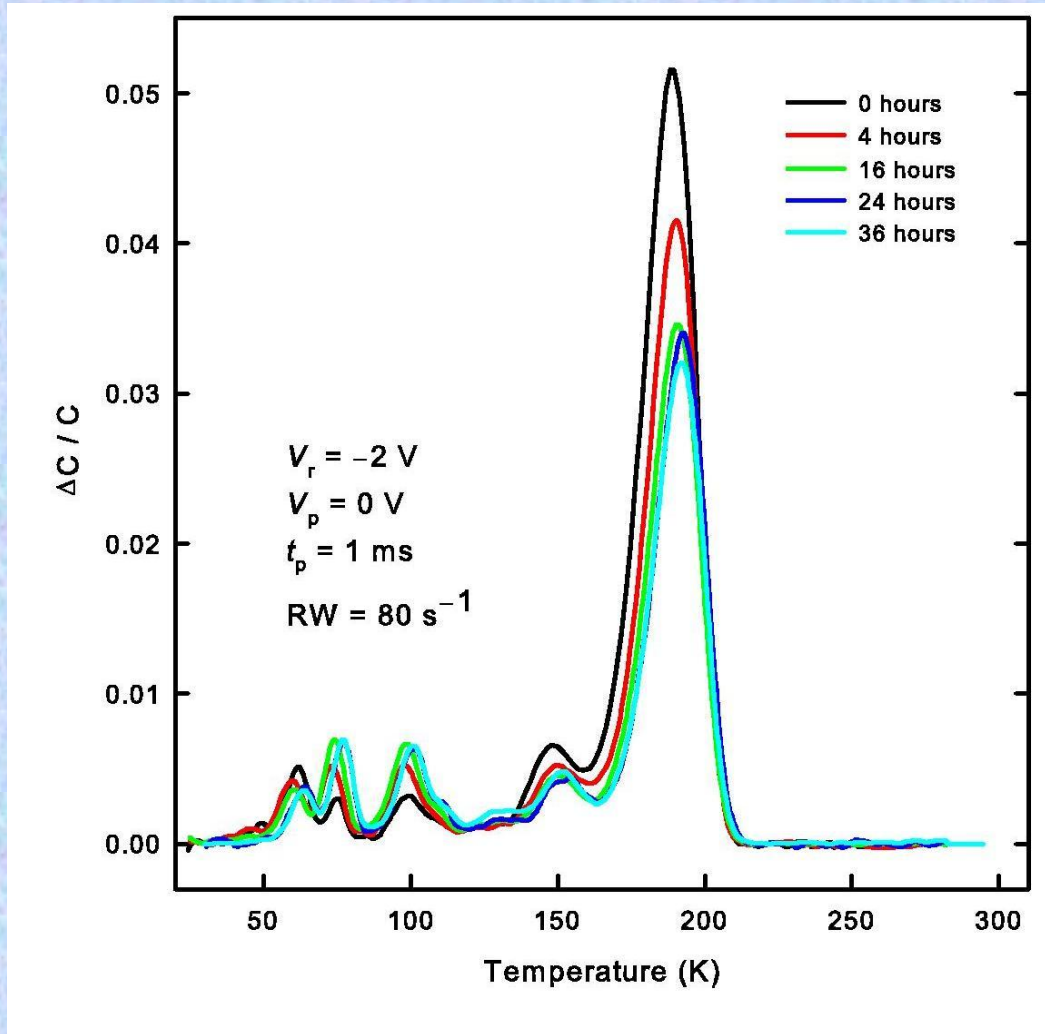
The profile  $N_t(x)$  can be obtained

Annealing: isochronal: same time, different temperatures

Time: annealing with time, same temperature.

**Annealing temperature and activation energy  $E_0$  can be obtained**

# Annealing at room temperature



$$\frac{dN_t}{dt} = -k(T)N_t$$

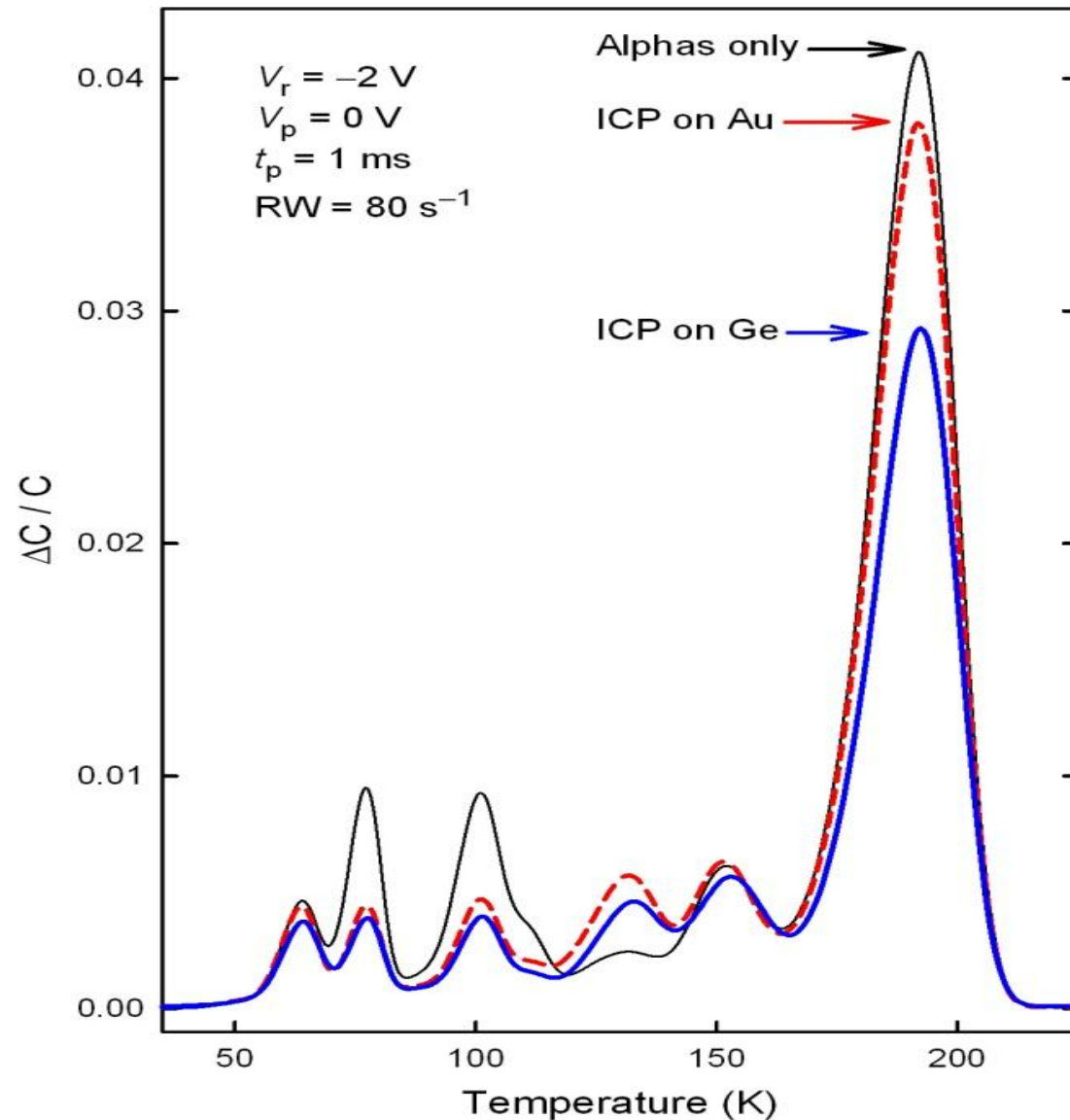
$$k(T) \propto \exp\left(-\frac{E_0}{k_B T}\right)$$

Annealing rate constant  $k(T)$  can be determined

**Isochronal annealing:** same time, different temperatures

Annealing activation energy  $E_0$  can be determined

# Our basic experiment: 4 eV ICP plasma annealing



1.-Sb doped Ge is damage with 5 MeV alpha particles

2.- Rest 24 hours

3.-Au diode is evaporated in half the sample (half A)

4.- DLST in A  
(black, alphas only)

5.-ICP in A and B

6.- Au diode in B

7.- DLTS in A (red-dashed)  
(red, ICP on Au)

8.- DLTS in B  
(blue, ICP on Ge)

# Our basic experiment: Facts

- 1.-Sb concentration:  $1.3 \cdot 10^{15} \text{ cm}^{-3}$  ( $n_i = 2.4 \cdot 10^{13} \text{ cm}^{-3}$ ) ; 1 Sb per  $10^8$  Ge
- 2.- Metal (Au) thickness: 25nm
- 3.- After ICP on Ge the E center concentration drops 29% from  $N_T = 1.07 \cdot 10^{14} \text{ cm}^{-3}$
- 4.- If ICP is done on Au, the E center reduction is smaller, but exists.
- 5.- ICP is done for 30' in intervals to prevent heating
- 6.- Defect annealing occurs up to 2600 nm or 4600 lattice units
- 7.- If the plasma energy is increased the effect is smaller
- 8.- Thermal annealing has to be done at 150 C to obtain a similar effect

# **Our hypothesis: Ar ions impacting on Ge produce Discrete Breathers, which travel through Ge and anneal the defects.**

## **Why?**

- 1.-DBs with MD in metals by Hyzhnyakov group have 0.5-5 eV
- 2.-The maximum energy transfer from Ar to Ge is 3.6 eV
- 3.- The activation energy for annealing an E center is about 1.36 eV
- 4.- The energy to anneal a defect has to remain localized up to  $10^4$  lattice units
- 5- The energy delivered by Ar atoms has to remain localized while traveling  $10^4$  lattice units
- 6.- Increasing the energy of the plasma does not enhance the effect, this is because DBs typically have a definite range of energies.
- 7.- At least stationary DB have obtained for Si and Ge with MD.

## Some numbers

1.-Ion current can be measured,  $\Phi_{Ar} = 1.25 \cdot 10^{13} \text{ cm}^{-2} \text{ s}^{-1} \approx 0.04 a^{-2} \text{ s}^{-1}$

2.- DB creation efficiency:  $\gamma: \Phi_{DB} = \gamma \cdot \Phi_{Ar} ; \gamma < 1$

3.-Number of breathers:  $n_{DB} = \frac{\Phi_{DB}}{c_s} \approx \gamma 2.3 \cdot 10^7 \text{ cm}^{-3}$

4.- Phonons:  $E_{ph} = 0.035 \text{ eV}$   $n_{ph} = \frac{3n_{Ge}}{\exp(E_{ph}/k_B T) - 1} \approx 4.6 \cdot 10^{22} \text{ cm}^{-3}$

$$\text{For } \gamma = 1 \quad -\frac{1}{N_T} \frac{dN_{T,DB}}{dt} \approx 2 \cdot 10^{-4} \text{ s}^{-1} \quad -\frac{1}{N_T} \frac{dN_{T,RT}}{dt} \approx 2 \cdot 10^{-11} \text{ s}^{-1}$$

Relative annealing efficiency per DB or phonon:  $\sim 10^{22}$

Relative annealing efficiency per eV of DB or phonons:  $\sim 10^{20}$

# Interaction cross-section and energy delivered by a breather

Interaction cross-section  $\sigma$

$$-\frac{dN_T}{dt} = \sigma N_T \Phi_{DB} \exp\left(-\frac{E_0 - \Delta}{k_B T}\right)$$

Interaction cross-section  $\sigma = \alpha \sigma_0$  ;  $\alpha > 1$

Minimal interaction cross-section  $\sigma_0 = n_{Ge}^{-2/3} \approx 10^{-15} \text{cm}^2$

Apparent diminution of the activation energy because of DB interaction:  $\Delta$

$$-\frac{dN_T}{dt} = \alpha \gamma \sigma_0 N_T \Phi_{ions} \exp\left(-\frac{E_0 - \Delta}{k_B T}\right) \quad \alpha, \gamma, \Delta \quad \text{unknown}$$

$$-\frac{dN_T}{dt} = \sigma_{ions} N_T \Phi_{ions} \quad \sigma_{ions} \approx \frac{1}{50} \sigma_0 \approx 1.5 \cdot 10^{-17} \text{cm}^2$$

For  $\alpha \gamma = 1$ ,  $\Delta \approx 1.2 \text{eV}$



## **Conclusions:**

0.- Plasma of 4eV produces annealing of defects very deep in Ge

## **Likely conclusions:**

1.- 4 eV Ar hits produce DB in Ge with very high efficiency

2.- DB of energy  $\sim 3\text{eV}$  travel distances of the order of at least  $10^4$  lattice units

3.- The annealing efficiency of DB with respect to phonons is extremely large

4.- The energy delivered by a DB to a defect is  $\sim 1.2\text{ eV}$

# References

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